Novel Solanum torvum Fruit Biomass-Derived Hierarchical Porous Carbon Nanosphere as Excellent Electrode Material for Enhanced Symmetric Supercapacitor Performance

by Rika Taslim

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2D MATERIALS - PREPARATION, PROPERTIES & APPLICATIONS

### Novel *Solanum torvum* Fruit Biomass-Derived Hierarchical Porous Carbon Nanosphere as Excellent Electrode Material for Enhanced Symmetric Supercapacitor Performance

ERMAN TAER,  $^1$  NURSYAFNI SYAMSUNAR,  $^1$  APRIWANDI APRIWANDI,  $^1$  and RIKA TASLIM  $^{2^{2,3}}$ 

1.—Department of Physics, Faculty of Mathematic and Natural Sciences, University of Riau, Pekanbaru 28293, Indonesia. 2.—Department of Industrial Engineering, State Islamic University Sultan Syarif Kasim Riau, Pekanbaru 28293, Indonesia. 3.—e-mail: rikataslim@gmail.com

Biomass-based hierarchical porous carbon nanospheres offer an outstanding performance of electrode materials in electrochemical energy storage device applications. However, integrating all these advantages into one fabric is still a challenge. Therefore, this study aims to develop novel biomass of *Solanum torvum* fruit (STF) as a hierarchical porous carbon nanosphere source for high-quality electrode material for supercapacitor applications. The STF-based carbon nanospheres were synthesized with a green, sustainable strategy through ZnCl<sub>2</sub> impregnation, carbonization, and physical activation. Through the 0.5 M ZnCl<sub>2</sub>, it was discovered that the carbon nanosphere maintains a dense spherical structure with enriched 3D "cow tripe-like" hierarchical pores. The optimized carbon nanosphere yielded a high specific surface area of 1176.29 m<sup>2</sup>g<sup>-1</sup> with a nearly balanced combination of the micro-mesopores. The combination of the 3D hierarchical pore structure and densely packed nanospheres gave high **673** trochemical properties of the symmetric supercapacitor with a delightful specific capacitance of 154 Fg<sup>-1</sup> at 1 Ag<sup>-1</sup> in the H<sub>2</sub>SO<sub>4</sub> electrolyte and high cyclic performance with coulombic efficiency ~ 84.5%. The energy density was boosted to 30.4 Whkg<sup>-1</sup> in power density of 1.27 kWkg<sup>-1</sup> 5 Ag<sup>-1</sup>. Therefore, porous carbon nanospheres from novel STF biomass are ideal candidates as electrode materials for high-performance electrochemical energy storage devices.

#### INTRODUCTION

In the last decade, electrochemical energy storage devices have been developed rapidly and applied in the industrial sector, such as batteries, fuel cells, and supercapacitors.<sup>1-4</sup> Batteries and fuel cells are considered not to meet the need for effective, efficient, and sustainable green energy storage due to their limited lifetime and toxic-corrosive residues, although lithium-ion batteries display different performances. Due to their good performance in terms of high power density and reversibility fast

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charge–discharge, and long cycle life,<sup>5–7</sup> supercapacitors have attracted much attention as highpower energy storage devices.<sup>8,9</sup> Supercapacitors are promising green energy storage that can be applied in various devices such as electric vehicles, smart power grids, dynamic braking, and important components of laser systems.<sup>10</sup> However, the relatively low energy density of supercapacitors hinders their practical application in energy storage.<sup>11,12</sup> This makes it necessary to develop strategies for increasing the energy density capability without reducing its power and life cycle.<sup>13</sup>

Electrode-based material is the main key to improving the performance of supercapacitors, especially in increasing their energy density. Several basic materials that have been investigated to be

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used as supercapacitor electrode materials include metal oxides, polymers, graphene,<sup>14,15</sup> and carbon nanotubes.<sup>16</sup> However, these materials have limitations because of the high production costs, complicated synthesis processes,<sup>17</sup> and harmful effects on the environment.<sup>18</sup> Previous studies also reported that carbon materials have a high potential as highquality electrode mating als due to their low cost, sustainability, good electrical conductivity, high porosity, promising chemical stability, and large specific surface area.<sup>19-23</sup> In addition, materials derived from biomass are relatively easy to obtain and abundantly available, and a sustainable-pollution-free approach makes them very environmentally benign. Furthermore, activated carbon has wider applications in air purification, water treatment, energy storage, and CO<sub>2</sub> capture.<sup>24</sup> This gains activated carbon gain much attention, especially as raw electrode materials, because they can significantly improve the electrochemical performance of supercapacitors.<sup>25</sup> The increase in surface area does not always support high-performance capacitive supercapacitors. This is due to the assumption of a micropore structure that contributes to the high surface area that is not fully accessible in the formation of the electrical double layer. The high surface area also allows the degradation of the electrical conductivity of the electrode material, which significantly affects the reduction of the high power of electrochemical energy storage devices.

Recent studies have suggested that the electrochemical performance enhancement is not only dependent on the high surface area, but also determined by the diverse pore size distribution (micro, meso, and macroporous), 2D nano morphology (nanosphere, nanosheet, and nanofiber), and heteroatom dopant (N, O, P, S, B).26 Therefore, optimizing the modification of the pore architecture is the best way to improve the supercapacitor performance. Modification of electrodes with interconnected hierarchical pore structures in the ranges of micro, meso, and macro is an option because it can increase the energy density of supercapacitors.<sup>27-29</sup> Micropores increase the specific surface area, mesopores reduce ion transport resistance, while macropores are similar to ion buffer reservoirs, which can shorten the ion diffusion dis-tance.<sup>30</sup> Moreover, hierarchical pore properties can be obtained in porous carbon based on biomass waste to support sustainable and environmentally benign energy storage technologies. A porous carbon with a hierarchical pore structure has been obtained by Zheng et al.<sup>31</sup> from kapok flower biomass waste precursor, which exhibits outstanding electrochemical performance with a specific capacitance of 286.8  $Fg^{-1}$ . The same result was also expressed by Xi et al.<sup>32</sup> who have converted cornstalk pith into 3D hierarchical porous carbon. It was discovered that the materials have increased the energy density of the supercapacitor by 13.4 Wh kg<sup>-1</sup> at a significant rate capability. This

hierarchical pore architecture was also confirmed in activated carbon from the precursors of alfalfa flower,<sup>33</sup> waste tea,<sup>34</sup> and jujube fruit.<sup>35</sup> However, the abundant 3D hierarchical pore behavior covering all surfaces of the electrode material can reduce their electrical conductivity, which limits the practical application of energy storage devices.

The 2D nano morphology with a uniform structure significantly affects the increase in the electrical conductivity of the electrode material. Nanospheres as one of the 2D nanoscales have gained significant attention as supercapacitor electrodes due to their distinctive shape with unique advantages such as uniform geometry, high density, and good conductivity.<sup>36</sup> According to Zheng et al. (2021), nanosphere structures have shown their tremendous potential in enhancing the performance of supercapacitors.<sup>37</sup> They have demonstrated a nanosphere morphology with an outstanding cyclical performance of  $\sim 96.5\%$ . Hollow microporous carbon nanospheres were obtained by simply treatment of KOH-activated pyrolyzing with a retention capacitance of 86.7%.<sup>38</sup> Nickel-cobalt-based nanospheres obtained through metal-organic framework (MOF) techniques have also confirmed their high performance with an outstanding energy density of 55.4 Wh kg<sup>-1</sup> at a cyclical performance of 98.8%. However, the original materials and methods applied have drawbacks. This is because the materials used are not environmentally friendly; they are corrosive, toxic, and expensive, followed by complicated, tedious, and time-consuming procedures, limiting their production on a large scale.<sup>3</sup> Meanwhile, biomass-based carbon nanospheres offer good density, evenly distributed pores, and defined particle size through synthesis techniques that are more environmentally friendly, sustainable, and free of toxic/corrosive residues. This is to provide a good ion transport pathway on the surface of carbon electrodes without reducing their high electrical conductivity.<sup>37</sup> Shallot peel biomass has recently been converted as carbon nanosphere through multiple activations without applying complicated techniques such as MOFs, templates, or strobe. The carbon nanofibers obtained have oxygen as a self-doping heteroatom function, which can increase the supercapacitor energy density by 16.7 Wh  $kg^{-1,41}$  The carbon nanosphere obtained from the biomass precursor corn straw has also been applied to the anode material for lithium-ion batteries.<sup>42</sup> This reveals the great potential of biomass as a carbon nanosphere source for electrochemical energy storage applications. This potential is still difficult to obtain and poses a serious challenge. This showed that there is a need to develop carbon nanofiber-based electrode materials from biomass materials.

Solanum torvum is a shrub that thrives in Indonesia, especially in Java and Sumatra islands. This plant is commonly known as Turkish berry and is widely distributed in tropical areas ranging from Novel Solanum torvum Fruit Biomass-Derived Hierarchical Porous Carbon Nanosphere as Excellent Electrode Material for Enhanced Symmetric Supercapacitor Performance

Indonesia, Malaysia, Philippines, Pakistan, India, tropical America, Africa, and the West Indies.<sup>43,44</sup> *Solanum torvum* fruits (STF) have a uniform spherical shape with a diameter ranging from 2 to 3 cm, which is rich in lignin and cellulose, enabling them to be a porous carbon source. The unique geometric shape also indicates their potential to present the nanosphere structure. Moreover, this is the first and most recent report on the potential of STF fruit as a carbon source with a nanosphere structure for supercapacitor applications.

In this study, activated carbon nanospheres with unique 3D "cow tripe-like" hierarchical pores were obtained from novel biomass of STF through an easy, green, environmentally benign, sustainable, and toxic-corrosive-free technique for electrochemical energy storage applications. STF was converted to porous carbon through a combination of precarbonization, chemical impregnation of ZnCl<sub>2</sub>, carbonization, and physical activation. The pore structure and surface morphology were controlled using activating agent concentrations at 0.3 M, 0.5 M, and 0.7 M, respectively. The optimized porous carbon displayed enriched nanosphere morphology, followed by an abundant hierarchical "cow tripelike" pore structure with a high specific surface area of  $1176.29 \text{ m}^2 \text{g}^{-1}$ . They also indicated the s71 doping of functional oxygen, thereby increasing the capacitive behavior of the electrode material. The best electrochemical prop 27 ies of supercapacitor cells were obtained with a specific capacitance of 154 F  $g^{-1}$  in 1 M H<sub>2</sub>SO<sub>4</sub> electrolyte in a twoelectrode system. Furthermor 18 its energy density was increased by 30.4 Wh kg<sup>-1</sup> at a current density of 1 A  $g^{-1}$ . These results indicated the great potential of STF as a source of hierarchical porous carbon nanospheres to improve the electrochemical properties of supercapacitor energy storage devices.

#### MATERIALS AND METHODS

#### Materials

Solanum torvum fruit (STF) was harvested from a plantation in Pekanbaru, Indonesia. Meanwhile, zinc chloride  $(ZnCl_2)$  for chemical activation was obtained from Emsure KgaA. Sulfuric acid  $(H_2SO_4)$  as an electrolyte was purchased from Panreac Quimica S.A.U. Distilled water was used to neutralize the sample and a 0.1 mm-thick duck eggshell membrane as a separator.

#### **Preparation of STF Materials**

STF was crushed upper until it became a slurry. The sample was dried using a doing oven at 110 °C and pre-carbonized through a vacuum oven at 250 °C for 2 h 30 min. Furthermore, the samples were produced by grinding using a mortar and milling. The sample powder from the milling was sieved using a 60  $\mu$ m sieve to obtain a uniformsize STF powder precursor.

#### **Preparation of Activated Carbon**

Activated carbon from STF was prepared by chemical activation using zinc chloride (ZnCl<sub>2</sub>) and one-step pyrolysis via carbonization and physical activation. The 30 g of precursor powder was mixed with ZnCl<sub>2</sub> solution at different concentrations such as 0.3 M, 0.5 M, and 0.7 M. The mixture was stirred using a hotplate at 80 °C with a rotation rate of 300 rpm. To facilitate data analysis, each sample was labeled with a code STF-3, STF-5, and STF-7 corresponding to the concentration of the activating agent applied. Subsequently, the chemically activated carbon sample was dried in a drying oven and printed using a hydraulic press with a pressure of 8 metric tons to obtain coins/pellets form. The pellets formed were  $\pm 2$  cm in diameter with a thickness of  $\pm$  0.2 cm. Subsequently, they were carbonized using  $N_2$  gas in a furnace from a temperature of 30°C-600°C, followed by physical activation characterized by the conversion of  $N_2$  gas to  $CO_2$  at 600°C, which started from a temperature of 600°C-800°C for 2 h 30 min. The coin-shaped carbon obtained from the pyrolysis results was cooled to room temperature and washed using distilled water. This washing was repeated for 5-6 days until the pH was neutral (7), and the washed carbon was dried for 24 h. In detail, preparation of hierarchical porous carbon nanospheres based on S. torvum fruit is shown in Fig. 1.

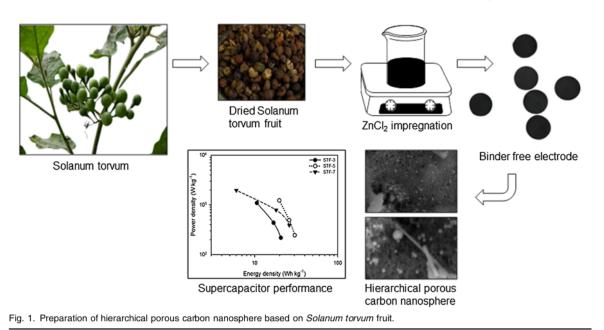
#### **Material Characterization**

The mass, diameter, and thickness were measured to determine the density of cathon pellets calculated through standard equations. The surface morphology of the samples was investigated by scanning electron microscopy (SEM) using the Zeiss Evo 10 i 68 rument. The phase and degree of crystallinity of the samples were examined using x-ray diffraction (XRD) techniques on X-Pert Powder Panalytical instruments with Cu K-a light sources in the range scattering angle  $2\theta$  at  $10^{\circ}-60^{\circ}$ . The porosity properties of STFs such as surface area and pore structure were reviewed through the N<sub>2</sub> gas absorption technique at 77 K us 45 Quantachrome Touchwin V1.22. The specific surface area was obtained from the Brunauer-Emmett-Teller (BET) method. Furthermore, the T-Plot method was used to calculate the pore volume and surface area of the micropores as well as for the pore size distribution obtained from the calculation of the Barrett-Joyner-Halenda (BJH) equation.

#### Supercapacitor Cell Performance Evaluation

The supercapacit (74 device consists of two electrodes made of  $\{31\}$ , a duck eggshell membrane as a separator, and 1 M H<sub>2</sub>SO<sub>4</sub> as an electrolyte. The performance of supercapacitor cells was evaluated by cyclic voltammetry (CV) and galvanostatic charge discharge (GCD) methods on a two-electrode

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48) figuration system. CV testing was carried out at scan rates of  $1 \text{ mV s}^{-1}$ ,  $2 \text{ mV s}^{-1}$ , and  $5 \text{ mV s}^{-1}$ using the CV UR Rad-ER 5841 physics tool. Subsequently, GCD testing was reviewed through the CD UR F28-ER 2018 instrument at a current density of 1 A at a scanning rate of 1  $1_{33}$  s<sup>-1</sup>. The electrochemical properties such as specific capacitance (Csp, F  $g^{-1}$ ), energy density (E, Wh  $kg^{-1}$ ), power density (P, W  $kg^{-1}$ ), and resistance (R,  $\Omega$ ) obtained were calculated using the standard formula discussed in the previous study.

#### RESULTS AND DISCUSSIONS

Activated carbon from STF was evaluated for density changes as shown in Fig. 2. Evaluation of density changes is an initial review to analyze the physical properties of activated carbon in form of solid coins/monolith without adhesive. From the initial treatment to powder prepared stage, STF exhibit carbon yield of 43%. Furthermore, the carbon powder is designed in the form of solid coins/monolith without adhesive and is followed by an integrated pyrolysis stage. Before a step of pyrolysis through carbonization and physical activation, the density of STF ranged from  $0.8912 \text{ g cm}^{-3}$ ,  $0.8863 \text{ g cm}^{-3}$ , to  $0.9204 \text{ g cm}^{-3}$ . After pyrolysis, the density of all samples based on concentration variations decreased with values of  $5211 \text{ g cm}^{-3}$ , 0.3608 g cm $^{-3}$ , and 0.5357 g cm $^{-3}$  in STF-3, STF-5, and STF-7 samples, respectively. In addition, activated carbons of STF-3, STF-5, and STF-7 obtained from the integrated pyrolysis stage exhibit carbon yields of 83.33%, 84.61%, and 80.59%, respectively. This is due to the influence of the carbonization process and physical activation

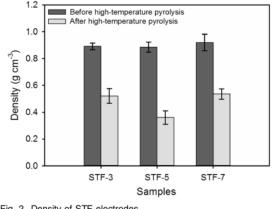


Fig. 2. Density of STF electrodes.

that causes the evaporation of volatile non-carbon compounds and reduces the dimensions of the monolith, including mass, volume, and density.4 The carbonization process in an N2 gas environment from 30 °C to 600 °C decomposes water, minerals, and all complex compounds including hemicellulose, cellulose, and lignin to reduce the volatile content, which yields a high carbon content.<sup>46</sup> The byproduct of the carbonization process in form of tar inhibits the carbon pores that are formed; therefore, a physical activation process is needed in a CO<sub>2</sub> gas environment.<sup>47</sup> The physical activation process from 600 °C to 800 °C aims to maximize tar reduction to form a better pore structure.48 The combination of these two processes can increase porosity, causing a decrease in sample density. The impregnation of the Novel Solanum torvum Fruit Biomass-Derived Hierarchical Porous Carbon Nanosphere as Excellent Electrode Material for Enhanced Symmetric Supercapacitor Performance

selected chemical activating agent significantly affects the dimensions of the coin solid carbon design. Increasing the concentration of ZnCl<sub>2</sub> from 0.3 to 0.5 M allows continuous etching of the carbon chain, which reduces the mass and volume of the sample precursor. This simultaneously presents abundant empty spaces improvising their porosity. Therefore, STF-5 showed 14 highest density reduction of 58.47%. However, an increase in the concentration of the activating agent to 0.7 M showed a decreased density reduction of about 41.71%. This is due to the excessive carbon chain etching that erodes the foundation of the precursor pore framework and cannot maintain the hierarchical pore structure, leading to the collapse of the various pore surfaces and covering the underlying pores.<sup>4</sup>

The novel biomass STF with a uniform spherical shape was converted to a porous activated carbon with a unique 2D nanosphere structure through the green, environmentally benign, and non-corrosivetoxic method. It was discovered that chemical activation using ZnCl<sub>2</sub> in 0.5 M solutions followed by high-temperature pyrolysis with N<sub>2</sub> and CO<sub>2</sub> gas eroded lignocellulosic compounds to present a unique 2D nanospherical shape. The morphology of the porous activated carbon nanospheres from STF was confirmed using scanning electron microscopy as shown in Fig. 3. The result showed that the STF-5-based carbon shows a highly interconnected 3D hierarchical pore morphological structure and uniform solid nanospheres, illustrating the unique "cow tripe-like" shape. Hierarchical pores have been confirmed in the mesoporous and macropores

ranges. Furthermore, Fig. 3a shows SEM images at 5000 magnification, revealing highly abundant nanospherical morphological structures on all precursor surfaces. The 2D nanospheres were found in all chunks, aggregates, and inner carbon blocks. In a larger area, the SEM images confirmed the rich hierarchical pores that are interconnected with each other resembling a unique "cow tripe-like" shape as shown in Fig. 3b. Furthermore, the nanospheres were evenly distributed in the hierarchical pores, which allow the porous carbon material to have high porosity properties and maintain good electrical conductivity for the supercapacitor electrodes. In this area, 2D nanospheres were found with diameters ranging from 162 to 519 nm, while the hierarchical pore morphology at the nanoscale was between 108 and 471 nm. In the high-temperature carbonization and pyrolysis processes, chemical activators hydrate and degrade complex lignocellulosic compounds, including lignins, that make important contributions in showing tubular structures and are speculated to present nanospheres on carbon-based biomass. The early reaction of ZnCl<sub>2</sub> impregnation also allows high exploitation of the precursor potential to completely etch the carbon chain, leading to a unique hierarchical "cow tripelike" pore structure. The shallow 3D interconnected pore shape allows the biomass carbon-based electrode materials to optimize the insertion/de-insertion of ionic charges in all directions to form the optimum electrical double layer without compromising their high power density. At 20,000 magnification, the nanospheres and the hierarchical pores

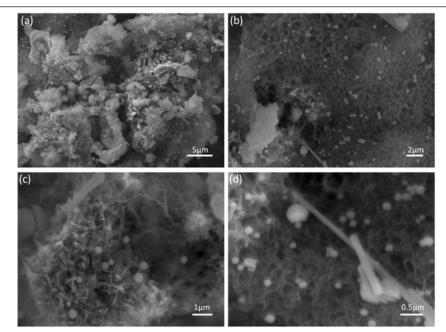


Fig. 3. (a-d) SEM image of hierarchical porous carbon nanosphere-derived Solanum torvum fruit (STF).

of "cow tripe-like" and rod-like structure are visible, as shown in Fig. 3c, with nanoseries size ranging from 270 to 969 nm, 184 to 467 nm, and 318 to 582 nm, respectively. Furthermore, the material has interconnected pores to form a bovine tripe pore structure, and an abundance of nanospheres contributes to enhancing the high porosity behavior and excellent electronic conductivity of the electrode material. This significantly increases the effectiveness of the electrode surface area in charge storage, and the interconnected pores can form a better electrode-electrolyte interface.<sup>50</sup> The existence of the structure can support high-performance electrode materials as a high-quality, environmentally benign, and pollution-free supercapacitor energy storage device.<sup>51</sup> A rod-like structure/fiber was also confirmed in Fig. 3d because the cellulose is being leached at high temperatures, which reduces its size.<sup>49</sup> The nanospheres at this magnification range in size from 205 to 423 nm, while the hierarchical pore sizes are from 65 to 561 nm. A deeper analysis is poured into the analysis of N2 gas absorption and electrochemical performance through CV and GCD.

The x-ray diffraction pattern of porous carbon based on novel STF biomass in different concentrations of activating agents is illustrated in Fig. 4. The x-ray diffraction pattern is the output generated for STF-3, STF-5, and STF-7 samples with an angle of  $2\theta$  in the range of  $10^{\circ}$  to  $60^{\circ}$ . All samples showed the presence of two broad peaks, each of which is located in a different hkl plane of 002 and 100 scattering planes. This broad peak occurred because of low crystallinity; therefore, the activated carbon electrode is amorphous.<sup>52</sup> The wide peaks in the scattering planes 002 and 100 with scattering angles ranging between 21°-25° and 42°-45° indicate that the porous carbon STF has good electrical conductivity.<sup>53</sup> This characteristic pattern of initiating abundant micropore structures was confirmed at the gently shifting peaks of the 002 scattering plane. The 002 scattering field shifts from 25 to 21 frequently with increasing concentrations of the

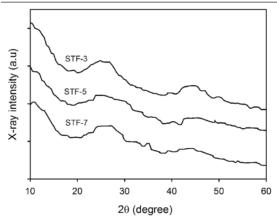


Fig. 4. XRD pattern of STF samples.

ZnCl<sub>2</sub> activating agent, which indicates the development of their narrower pores due to the reaction of carbon with an intense activating agent as shown in Table II. The behavior of the rich micropores is required to provide suitable ionic charge sites for accessibility at the electrode/electrolyte interface. This characteristic XRD pattern is similar to that of several previous investigations, which also showed similar x-ray diffraction characteristics such as activated carbon from tea waste biomass precursors,<sup>54</sup> garden waste,<sup>17</sup> neem leaves,<sup>55</sup> bark waste potatoes,<sup>56</sup> and argan shells<sup>57</sup>.Table I shows the parameter data of the carbon lattice of S. torvum fruit, which experienced a shift in the scattering angle causing changes in the value of layer height  $(L_c)$  and layer width  $(L_a)$  and showing the distance between the planes  $(d_{002})$  and  $(d_{100})$ . According to the Bragg  $2d\sin\theta = n\lambda$  equation, the average distance between layers  $d_{002}$  and  $d_{100}$  ranged from 0.348 nm to 0.419 nm and 0.200 nm to 0.204 nm, respectively. The d<sub>002</sub> value obtained was higher than the normal graphite  $d_{002}$ ; this indicated that STF carbon has weak graphite properties with better amorphous properties, which initiate the formation of high porosity. The value din he layer height  $(L_c)$  is related to the prediction of the specific surface area of the carbon material according to the empirical equation SSA =  $2/\rho_{xrd}L_c$ . The SSA value obtained for the STF-3 sample is 1230 m<sup>2</sup> g<sup>-1</sup>, while for the STF-5 and STF-7, the values are  $2244 \text{ m}^2 \text{ g}^{-1}$ and 1830 m<sup>2</sup> g<sup>-1</sup>, respectively. This indicated that a small  $L_c$  value can support the formation of an active site suitable for ion diffusion at the electrolyte/electrode interface.

The porosity properties of 2D caran nanospheres based on STF were evaluated using Brunauer Emmett Teller (BET), t-Plot, and Barrett Joyner Halenda (BJH) methods to obtain specific and microporous surface area, as well as pore size dist 76 ution. Figure 5a and b illustrates the nitrogen adsorption-desorption profiles and the pore size distribution of STF-3, STF-5, and STF-7. Moreover, the characterization results for STF showed a type I and IV collaboration isotherm curve with an H4type hysteresis loop.<sup>59</sup> At relatively low-pressure, P/ P0 < 0.1 indicates an increase in the absorption volume due to the presence of abundant micropores, which continue until a higher pressure of 0.4 < P/P0 < 0.91. This is followed by an H4 hysteresis loop, indicating the presence of a well-developed mesopore. At relatively high pressures, a slightly increased slope confirms the presence of macropores.<sup>60,61</sup> This indicated that a hero pore structure with high porosity was obtained by chemical activation of ZnCl<sub>2</sub> in the STF biomass, as shown in Fig. 3. Furthermore, the difference in concentration of ZnCl<sub>2</sub> 8 STF-activated carbon had a significant effect on the specific surface area and pore volume. The addition of an activator concentration of 0.3 M to 0.5 M increased the surface area and pore volume as presented in Table II. In STF-3, the surface area

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| Sample | $\frac{2	heta_{002}}{2	heta_{002}}$ (°) | $\frac{2\theta_{001}}{2}$ (°) | $d_{002}$ (Å) | <b>d</b> <sub>100</sub> (Å) | $\mathbf{L}_{\mathbf{c}}$ (Å) | $\mathbf{L}_{\mathbf{a}}$ (Å) |
|--------|---|-------------------------------|---------------|-----------------------------|-------------------------------|-------------------------------|
| STF-3  | 25.512                                  | 45.172                        | 3.488         | 2.005                       | 7.455                         | 27.909                        |
| STF-5  | 23.272                                  | 44.304                        | 3.819         | 2.042                       | 5.691                         | 7.822                         |
| STF-7  | 21.145                                  | 44.350                        | 4.198         | 2.040                       | 6.030                         | 12.834                        |

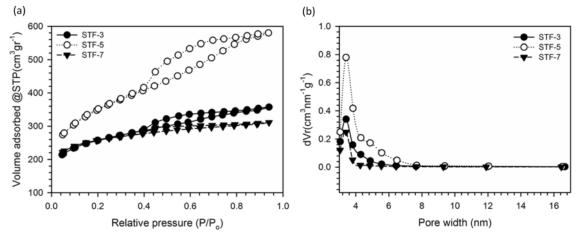


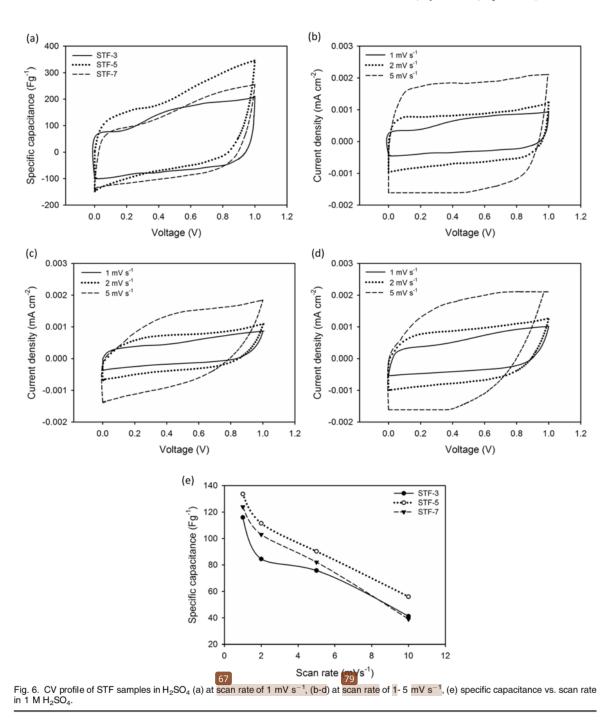
Fig. 5. (a) Nitrogen adsorption-desorption profiles of STF samples and (b) pore size distribution of STF samples.

| Table II. | Porosity | properties | of STF | samples |
|-----------|----------|------------|--------|---------|
|-----------|----------|------------|--------|---------|

| Sampel | $\frac{{{S_{BET}}\left( {{m^2}\;{g^{ - 1}}} \right)}}{{}$ | ${\bf S_{micro}}~(m^2~g^{-1})$ | ${{\bf S}_{meso}}~(m^2~g^{-1})$ | $\frac{\mathbf{v_{tot}} \ (\mathbf{cm^3} \ \mathbf{g^{-1}})}{\mathbf{cm^3}}$ | $D_{aver} (cm^3 g^{-1})$ |
|--------|---|--------------------------------|---------------------------------|--|--------------------------|
| STF-3  | 839.959   | 535.526                        | 304.433                         | 0.554049   | 1.31923                  |
| STF-5  | 1176.29   | 544.006                        | 632.289                         | 0.899726   | 1.52976                  |
| STF-7  | 821.09  | 628.164                        | 192.926                         | 0.482467   | 1.17519                  |

value was  $839,959 \text{ m}^2 \text{g}^{-1}$  with a pore volume 38 0.554049 cm<sup>3</sup> g<sup>-1</sup> and increased in STF-5 with a surface area value of 1176.29 m<sup>2</sup> g<sup>-1</sup> and the pore volume of 0.899726 cm<sup>3</sup> g<sup>-1</sup>. This increase occurred because of the ZnCl<sub>2</sub> activator, which acts as a reducing agent and breaks the lateral bonds in the cellulose molecule to produce activated carbon with a high surface area. STF-5 also shows significant mesoporous productivity, as presented in Table II. This nearly balanced combination of micro- and mesopores is very advantageous for optimizing the high performance of electrode materials in producing high energy density and power in supercapacitor electrochemical energy storage devices. The pore size distribution in the carbon nanosphere is illustrated in Fig. 5b. STF-5 shows an abundant mesopore size distribution compared to STF-3 and STF-7. This reinforces the presence of hierarchical pores in STF-5, supporting the high performance of their electrodes. However, at STF-7, the specific

surface area was reduced to  $821.09 \text{ m}^2 \text{ g}^{-1}$  and the total pore volume was  $0.482467 \text{ cm}^3 \text{ g}^{-1}$ . This decrease occurred because of damage to the hierarchical pore framework because of excessive etching of carbon chains by ZnCl<sub>2</sub> at a concentration of 0.7 M for the pores that had been formed to collapse and cover the well-formed pores underneath.<sup>62</sup> On the plus side, the more carbon chain etching on ST-7 allowed the precursor to having higher micropores than STF-3 and STF-5, as presented in Table II. The predominant micropores can support electrochemical performance with high surface area contributed and energy densities through maximum ion absorption.<sup>63</sup> The availability of a large number of mesopores also provides benefits for rapid ion transfer, which favors enhanced electrochemical performance.<sup>30</sup> This showed that the combination of pores can improve the performance of high storage applications, especially energy and power density.64 Therefore, it is concluded that the high surface



area with elevated pore volume and hierarchical porosity can facilitate easy ion transport and charge storage for prepared STF-5 to have a high potential for electrode materials in supercapacitor applications.<sup>65</sup>

The electrochemical performance of the STFbased carbon nanospigate for supercapacitor cell was evaluated using cyclic voltammetry (CV) and galvanostatic charge discharge (GCD) methods in a two-electrode configuration immersed in 1 M H<sub>2</sub>SO<sub>4</sub>

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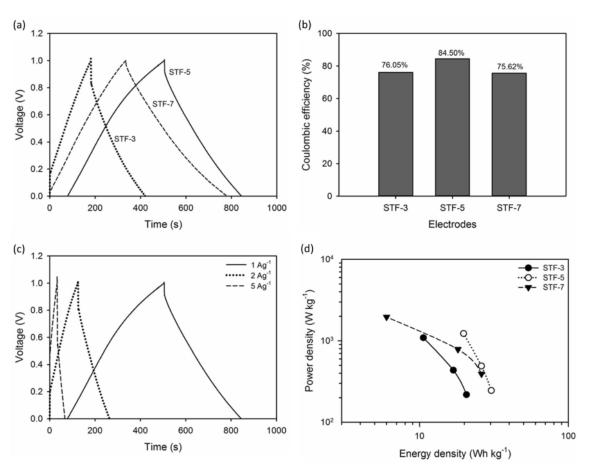
Novel Solanum torvum Fruit Biomass-Derived Hierarchical Porous Carbon Nanosphere as Excellent Electrode Material for Enhanced Symmetric Supercapacitor Performance

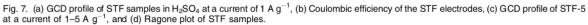
electrolyte. Figure 6a sh 23s the CV curves of the STF-3, STF-5, and STF-7 electrodes at a scan rate of  $1 \text{ mV s}^{-1}$  with potential intervals from 0.0 to 1.0 V. The CV curve shows a distorted rectangular shape characterizing the ideal electrical double-layer capacitance behavior for biomass-based activated carbon nanosphere electrodes.<sup>66,67</sup> The STF electrode also shows a curve with a fast spenific capacitance response at low voltages < 0.08 V. This is due to the fast diffusion of ions in the electrolyte towards the electrode surface, where the rich micropores are filled with ions to form a charge layer on electrode/electrolyte interface. At voltthe ages > 0.1 V, the capacitance response begins to subside, because the ion charge has filled the larger pores until the maximum voltage reaches 1.0 V. The STF-3 displays a rectangular shape followed by a potential hump of 0.2-0.8 V confirming the presence of pseudocapacitance effects derived from faradaic redox reactions. Functional oxygen contributes as a self-doping heteroatom, which enables a pseudocapacitance effect on the STF-3 electrode. However, STF-3 shows the smallest repart shape indicating low material properties to improve the electrochemical properties of the supercapacitor electrodes. The STF-5 electrode shows a wider rectangular shape compared to STF-3 and STF-7, confirming the high double-layer capacitance.<sup>52</sup> The addition of the activating agent concentration from 0.3 M to 0.5 M significantly increases the specific surface area. This allows the provision of multiple active sites in the STF-5 electrode, thereby forming an abundant electrochemical double layer. The high mesoporosity contributes to the smooth flow of ionic charges at the electrode/electrolyte interface, thereby increasing the high performance of the STF-5 electrode. The addition of higher concentrations can reduce the wettability of the samples through the evaporation of oxygen in form of  $CO_2$ and H<sub>2</sub>O, as confirmed in the CV profile of STF-5, which reduced their pseudocapacitance effect. A further increase in concentration up to 0.7 M shows a smaller rectangular shape than STF-5. This is because the STF-7 experienced a decrease in the specific surface area as discussed in the  $N_2$  gas absorption analysis, thereby reducing their high electrochemical properties. Therefore, it can be concluded that STF-5 has the highest capacitive behavior, followed by STF-7 and STF-3. The high performance of the STF 60 lectrodes was also observed at high sintering rates from  $1 \frac{57}{57} s^{-1}$  to  $5 \text{ mV s}^{-1}$ , as shown in Fig. 6b, c and d. At a high scan rate of  $5 \text{ mV s}^{-1}$ , the CV profiles of STF-3, STF-5, and STF-7 retaining a rectangular shape exhibit better EDLC properties.<sup>67</sup> The pseudocapacitance effect of functional oxygen as self-doping oxygen was degraded, revealing weak faradaic redox reactions in STF carbon nanosphere-based electrode materials. The capacitive properties of STF electrodes at different scanning rates are illustrated in Fig. 6e. It was discovered that the

specific capacitance decreases as the scanning rate increases because of their unbalanced pore distribution, which impedes electrolyte ion transport at high scanning rates.<sup>69</sup> This is called the diffusion limit, where it is difficult for electrolyte ions to diffuse into the pores of the electrode. There is an ineffective interaction between the electrode material and the electrolyte for the specific capacitance to decrease at high scanning rates.<sup>70</sup> The STF-5 electrodes show good cyclical performance by m<sub>23</sub>-taining their specific capacitance of about 56% at a scan rate of 5 mV s<sup>-1</sup>.

The electrochemical properties of STF carbon nanosphere electrodes were evaluated using galvanostatic charge-discharge (GCD), especially for specific capacitance, power density, and energy density. The GCD profile for an STF-based carbon nanosphere electrode immersed in 1 M H<sub>2</sub>SO<sub>4</sub> electrolyte at a current of 1 A  $g^{-1}$  is shown in Fig. 7a. The GCD profile revealed an imperfect isosceles triangle shape that indicated the characteristic electrical double-layer properties.<sup>71</sup> This is in line with the CV profile shown in Fig. 6 where the type of distorted isosceles triangle showed the reversibility of different ion charge-discharge times for each STF due to ion degradation caused by self-heteroatom, which shows a pseudocapacitance effect. The functional oxygen in form of OH/COOH on the electrode surface allows the addition of wettability properties to initiate faradaic redox reactions in the charging process. It is characterized by a convex curvature of the GCD profile. The low IR drop confirmed the STF electrode resistance of 0.155  $\Omega$ ,  $0.069 \Omega$ , and  $0.027 \Omega$  for the STF-3, STF-5, and STF-7 electrodes. Lower resistance indicates higher conductivity along with the addition of ZnCl<sub>2</sub>, which can increase the conductivity of the electrode.

The length of charging and discharging time indicates the high capacitive nature of the STF electrode.<sup>72</sup> In this study, STF-3 has a shorter charg 27 nd discharge time, showing that the electrode has a smaller specific capacitance of 118 F g The specific capacitance increased at STF-5, which was characterized by a longer charge and discharge time with a specific capacitance of 154  $Fg^{-1}$ . This increase was influenced by the ZnCl<sub>2</sub> ac 56 ator, which significantly increased the specific surface area from 839,959  $m^2 g^{-1}$  to 1176.29  $m^2 g^{-1}$ . The combination of 2D nanospheres and the unique 3D interconnected shallow "cow tripe" pore shape also allows the biomass carbon-based electrode material to have unique advantages such as uniform geometry, good density, well-distributed porosity, and well-defined particle sizes to optimize the insertion/ de-insertion of ionic charges in all directions from the optimum electrical double layer without compromising their high power density.73 The addition 58 the activating agent at 0.7 M led to a reduction in the specific surface area and a widening of the pore size, thereby reducing the specific capacitance by 14.93% to 131 F  $g^{-1}$ . The coulombic efficiency of the





STF electrode was also implemented through the ratio between charge and discharge time on the GCD curve, as shown in Fig. 7b. To is shows that the coulombic efficiency obtained for STF-3, STF-5, and STF-7 were 76.05%, 84.50%, and 75.62%, respectively. This proves that the STF-5 electrode has the best electrochemical performance compared to the STF-based carbon nanosphere source. The high electrochemical confirmations of the STF-5 electrode was examined by different current densities from 1 A  $g^{-1}$  to 5 A  $g^{-1}$ , as shown in Fig. 7c. Based on Fig. 7c, the GCD profile still maintains the distorted isosceles triangle shape, revealing good electrical double-layer properties. The STF-5 electrode shows that the charge-discharge time confirms the charge storage phenomenon, where the high current density hinders the electrolyte ions from having sufficient time to diffuse to complete the faradic reaction at the electrode surface.<sup>74</sup> This causes the electrochemical performance to reduce for the capacitance to decrease as the current density increases. The STF-5 electrodes maintained their high specific capacitance of 101 F  $g^{-1}$  at 5 A  $g^{-1}$  in the two-electrode configuration system.

The energy and power densities of the STF carbon nanosphere electrode were fully evaluated as shown in the second power density of 30.43 Wh kg<sup>-1</sup> with a maximum power density of 1.27 kW kg<sup>-1</sup> at a current density of 5 A g<sup>-1</sup>. Meanwhile, the STF-3 and STF-7 electrodes have lower energy densities of 26 Wh kg<sup>-1</sup> and 19 Wh kg<sup>-1</sup> at power densities of 1.10 k and 800 W kg<sup>-1</sup>, respectively. These results were compared with other biomass-based carbon sources, as shown in Table III.

#### CONCLUSION

A simple, green, sustainable, and corrosion/toxinfree synthesis route was demonstrated to obtain carbon nanosphere with a unique hierarchical pore "cow tripe-like" structure from the novel biomass of

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#### Table III. Comparison of the electrochemical performance of carbon biomass-based for supercapacitor electrodes

| Sources                                   | Morphological<br>structure                      | $\substack{S_{BET} \\ (m^2g^{-1})}$ | Electrode<br>type | Electrolyte                               | $\begin{matrix} E_{sp} \\ (Wh \\ kg^{-1}) \end{matrix}$ | $\frac{P_{sp}\left(W\right.}{kg^{-1}}$ | References |
|---|---|-------------------------------------|-------------------|---|---|--|------------|
| Camellia japonica flowers                 | Interconnected<br>spherical                     | -                                   | 21<br>3-electrode | 26<br>1 M KOH                             | 34.54   | 1600                                   | 75         |
| Tea saponin                               | Sheet   | 1550.3                              | 2-electrode       | 1 M<br>TEABF4                             | 27.01   | 1500                                   | 53         |
| Chitin                                    | Hierarchical por-<br>ous                        | 1158.69                             | 3-electrode       | 6 M KOH                                   | 22.07   | 980                                    | 76         |
| Seaweed powder                            | 3D honeycomb                                    | 1206.97                             | 3-electrode       | 6 M KOH                                   | 3.125   | 5000                                   | 77         |
| Camphor leaves                            | Hierarchical por-<br>ous                        | 2794                                | 3-electrode       | 6 M KOH                                   | 32.9  | 9461                                   | 78         |
| Lemon peel                                | Sheet   | _                                   | 3-electrode       | $1 \mathrm{M} \mathrm{H}_2 \mathrm{SO}_4$ | 11.84   | 361.8                                  | 79         |
| Foxtail grass seeds                       | Hierarchical por-<br>ous                        | 1428                                | 3-electrode       | 6 M KOH                                   | 18.2  | -                                      | 80         |
| Wheat husk                                | Hollow tunnels                                  | 1200                                | 3-electrode       | 6 M KOH                                   | _   | _                                      | 81         |
| Onion peel                                | Hierarchical por-<br>ous                        | -                                   | 3-electrode       | $1 \mathrm{M} \mathrm{H}_2 \mathrm{SO}_4$ | 13.61   | 200.8                                  | 82         |
| <i>Lessonia</i> trabeculata<br>macroalgae | Highly coupled<br>flake                         | 769                                 | 2-electrode       | 1 M <u>KOH</u>                            | 2.82  | 491                                    | 83         |
| Solanum torvum fruit                      | Hierarchical por-<br>ous carbon nano-<br>sphere | 1176.29                             | 2-electrode       | $1 \mathrm{M} \mathrm{H}_2 \mathrm{SO}_4$ | 30.4  | 1270                                   | This work  |

STF for high-performance supercapacitors. The pore structure, morphology, and electrochemical properties of activated carbon nanospheres were reviewed with the application of three different concentrations of ZnCl<sub>2</sub> activating agent. The STF-5 sample significantly confirmed the extraordinary potential of STF biomass to obtain activated carbon with a combination of rich nanosphere structure and unique "cow tripe-like" hierarchical pores. Furthermore, the impregnation of ZnCl<sub>2</sub> at hightemperature pyrolysis also increased the porosity of the high carbon nanosphere by 1176.29 m<sup>2</sup> g<sup>-1</sup> with a very rational micro:meso distribution. The symmetrical supercapacito 40 ssembled based on STF-3 electrodes exhibited a specific capacitance of 154 F g<sup>-1</sup> and an increased energy density output of 30.4  $^{63}$  kg<sup>-1</sup> at 1 A g<sup>-1</sup> in aqueous electrolyte H<sub>2</sub>SO<sub>4</sub>. Based on this result, it can be concluded that the novel biomass optimized through the proposed strategy is highly efficient and effective to obtain activated carbon with hierarchical porous nanosphere regulation as a promising electrode material for high-performance supercapacitor applications.

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#### CONFLICT OF INTEREST

The authors declare that they have no conflict of interest.

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